## Research Paper

# Controllable Growth of Single Layer MoS<sub>2</sub> and Resistance Switching Effect in Polymer/MoS<sub>2</sub> Structure

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**Abstract** We report a chemical vapor deposition approach and optimized growth condition to the synthesis of single layer molybdenum disulfide ( $MoS_2$ ). Obtaining large grain size with continuous  $MoS_2$  atomically thin films is highly responsible to the growth distance between molybdenum trioxide source and receiving silicon substrate. Experimental results indicate that triangular shape  $MoS_2$  grain size could be enlarged up to > 80 um with the precisely controlled the source-to-substrate distance under 7.5 mm. Furthermore, we demonstrate fabrication of a memory device by employing poly(methyl methacrylate) (PMMA) as insulating layer. The fabricated devices have a PMMA-MoS<sub>2</sub>/metal configuration and exhibit a bistable resistance switching behavior with high/low-current ratio around  $10^3$ .

**Keywords:** MoS<sub>2</sub>, CVD, Controllable grain size, Memory

#### I. Introduction

After discovery of grapheme, transition metal dichalcogenides(TMDs) have received much attention due to their interesting electronic and optical properties for application to novel devices [1-3]. One of the most investigated material, molybdenum disulfide (MoS<sub>2</sub>) present a direct band gap (~1.8 eV) in the monolayer form and revealed an indirect band gap (~1.2 eV) in the bulk state which makes them promising for future logic device beyond the grapheme. The systems for synthesizing MoS<sub>2</sub> have been widely examined in the past few years. Chemical vapor deposition (CVD) approach has been proved as a confident system in synthesizing large area and high quality single layer TMD layer [8-15]. The common technique for growing MoS2 layer was deposition of molybdenum trioxide (MoO<sub>3</sub>) or Mo on a silicon dioxide (SiO<sub>2</sub>) substrate and followed by sulfurization by a sulfur vapor at high temperature. The MoS<sub>2</sub> grown by sulfurization method usually has high surface coverage and good uniformity. However, the small grain size of the epi-layer is the major disadvantage. For the certain device application, large grain size and high surface coverage of the synthetic layer is the most important.

In this work, we have successfully synthesized  $MoS_2$  thin films with the thickness from single layer to few layers by using CVD system. In order to increasing grain size of single layer  $MoS_2$  the growth distance between substrate

and sources (such as MoO<sub>3</sub>) was optimized. Furthermore, the CVD grown MoS<sub>2</sub> thin film is integrated into resistance switching memory, while MoS<sub>2</sub> layer in this device takes on a role as a charge trapping and detrapping layer with dielectric materials encapsulation.

#### **II. Experiments**

CVD synthesis of MoS<sub>2</sub> was performed by using 99.999% purity MoO<sub>3</sub> powder and 99.999% purity sulfur (S) powder as the reactant materials. The concentration of the S vapor must be much higher than the MoO<sub>3</sub> precursor to complete the sulfurization. To accomplish this, 10 mg MoO<sub>3</sub> powder was put in the alumina boat, and the SiO<sub>2</sub> (270 nm-thick)/Si substrate ( $\sim$ 1.5 × 1.5 cm<sup>2</sup>) was placed on the alumina boat. The alumina boat was placed at the center of the heating zone of a 2-inch quartz tube. A total of 100 mg S of powder was supplied in another alumina boat and was located upstream relative to the carrier gas flow direction from the MoO<sub>3</sub> source. The heating zone was heated up to a temperature of 690°C at a heating rate of 16°C/min under argon atmosphere under a 100-sccm flow rate. Argon gas played as the carrier gas to make the diffusion of S vapor to the heating zone. The gas flow rate was set to be constant at 10-sccm after 690°C. When the growing procedure starts, the heating zone temperature was retained at 690°C for 5 minutes. The MoO<sub>3</sub> sources formed MoO<sub>3-x</sub> and further reacted to the S vapor to synthesis MoS<sub>2</sub> film at 690°C. The single layer MoS<sub>2</sub> thin film was grown by diffusing these sub-oxide compounds to the substrate. For the fabrication of the memory devices,

\*Corresponding author E-mail: ek-kim@hanyang.ac.kr synthesized MoS<sub>2</sub> film was coated with poly(methyl methacrylate) (PMMA) solution using spin coater. The PMMA films acted as both of the supporting layer for the transfer process and dielectric materials in the device. The PMMA-MoS<sub>2</sub> stack was further transferred onto a silicon substrate in which a gold film coated. A top aluminum electrodes (100 nm) were deposited by using thermal evaporator on PMMA-MoS2 layer to form Al-PMMA-MoS<sub>2</sub>-Au configuration.

To characterize the structure of MoS<sub>2</sub> film, Raman measurement was conducted with laser excitation energy of 532 nm. X-ray photoelectron spectroscopy (XPS) was used for examining the chemical composition, crystal structure, and residual elements. An optic microscopy was employed to determine the grain size and thickness. Current-voltage (I-V) characteristic measurements were carried out by using a semiconductor parameter analyzer.

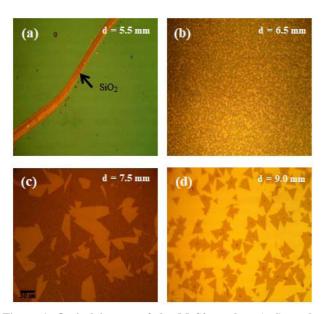


Figure 1. Optical images of the MoS2 product (a-d) at the difference growth substrate.

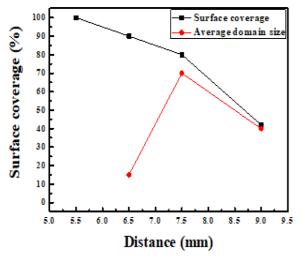


Figure 2. Surface coverage and average domain size at difference distance from source to substrate.

## III. Result and Discussion

Controlling the layer number and grain size of the CVDgrown MoS<sub>2</sub> layer is achieved by adjusting the distance from MoO<sub>3</sub> sources to substrate. Figure 1(a-d) shows the optical image of the CVD-grown MoS2 at various distances from 5.5 to 9.0 mm. Figure 1(a) shows the typical few layer MoS<sub>2</sub> film grown, the substrate is fully covered by  $MoS_2$  thin film. For the short distance (~5.5 mm), the high concentration of the MoO<sub>3</sub> gaseous deposited on the substrate directly. The higher gaseous MoS<sub>2</sub> assured that the much chemical reaction occurred with the relatively short distance. When increasing the distance to be 6.5 mm, The triangular shape MoS<sub>2</sub> fully cover the growth substrate as shown in Fig. 1(b). In this distance, the space is not allowed for growing the large grain. The average grain size is around 10 µm as increase the distance between sources to substrate. Figure 1(c) shows the resulting MoS<sub>2</sub> layer when the distance set to be 7.5 mm. The overall surface coverage has been reduced from 90 to 85%, but the average grain size has increased from 10 um to 80 um. The triangular shapes are presented at the edge of substrate where nuclei density is relatively low. It also means enough space is essential for growing the large grain size of MoS<sub>2</sub>. Some overlying triangular shape flakes embody the original process of the formation of the MoS<sub>2</sub> thin film. It is decent displaying that the large grain size MoS<sub>2</sub> flake (> 80 µm) has been well synthesized at this distance. As the distance increases more than 7.5 mm, the overall surface coverage is decreased to 45% and the average domain size also decreased to 40 µm (Fig. 1(d)). The low concentration of gaseous MoS<sub>2</sub> failed to supply enough sources gas, causing the lower surface coverage and small grain size. Figure 2 shows the average domain size and the surface coverage as a function of distance from source to growth substrate. The surface coverage accommodates 100% at the 5.5 mm distance and slightly decreases as increasing the distance. This is because that the high gaseous MoS<sub>2</sub> provides both high-density nuclei and plentiful source gas supply. Plentiful gaseous MoS2 insures that the highdensity nuclei can keep synthesizing until the grain merge each other. On the other hand, the moderate low-density nuclei and gaseous help to synthesize large grain size

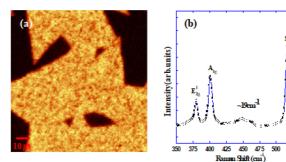


Figure 3. (a) Raman mapping image of MoS<sub>2</sub> grown by CVD method, (b) Raman spectra of MoS2 grown by CVD method.

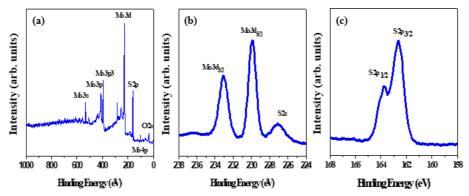


Figure 4. (a) XPS survey scans for CVD-MoS<sub>2</sub> and (b) the XPS spectra of the Mo 3d state, (c) S 2s state.

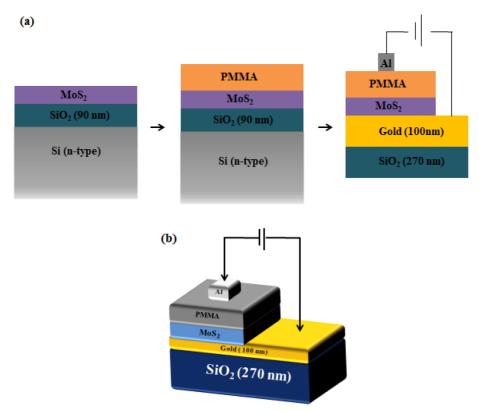


Figure 5. (a) Transfer process of the CVD-MoS<sub>2</sub> for the fabrication of memory device. (b) Schematic diagram of the PMMA/ MoS<sub>2</sub>/gold structure as a memory device.

without decreasing the surface coverage. However, in relatively long distance (more than 7.5 mm), the average domain size and surface coverage both reduced.

Figure 3(a) and (b) show the Raman mapping image of MoS<sub>2</sub> and Raman spectra of the CVD-grown MoS<sub>2</sub>, respectively. It shows that MoS<sub>2</sub> with around 80 um of grain size was synthesized on the substrate. Raman spectra demonstrated two characteristic peaks at 402.27 and 383.18 cm<sup>-1</sup>, which are  $A_{1g}$  and  $E_{2g}^1$  modes, respectively. The E<sup>1</sup><sub>2g</sub> mode is correlated with the vibration of the Mo-S bond along the basal plane. While the  $A_{1g}$  mode is due to the vibration of the sulfur atoms along the vertical axis. The difference between the  $E^{1}_{2g}$  and  $A_{1g}$  peak was used to confirm the thickness of the MoS2 layer, and it was found to be 19.09 cm<sup>-1</sup>, corresponding to a single layer structure [16]. The full width at half maximum (FWHM) of the  $E^{1}_{2g}$ 

peak is known as an indicator for crystalline quality [17]. The FWHM of  $E^{1}_{2g}$  peak from the synthesized MoS<sub>2</sub> is 4.35 cm<sup>-1</sup> which was correlated to the high quality crystalline of MoS<sub>2</sub>.

The surface chemistry of the CVD-grown MoS<sub>2</sub> was characterized by XPS measurements as shown in Fig. 4. From the survey scans of Fig 4(a), the Mo, S and O peaks can be found. In Fig. 4(b), the XPS spectra shows two peaks at 229.7 and 233.6 eV, corresponding to the Mo 3d <sub>3/2</sub> and Mo 3d<sub>5/2</sub> state, respectively. Fig. 4(c) shows the binding energies of S  $2p_{1/2}$  and S  $2p_{3/2}$  at 163.16 and 162.53 eV, respectively. The 2H-MoS<sub>2</sub> crystal for Mo 3d shows two peaks at 229.7 and 233.6 eV that can be attributed to the doublet Mo  $3d_{5/2}$  and  $3d_{3/2}$  [18]. The data of 229.3 and 233.2 eV from the CVD-grown MoS<sub>2</sub> are in agreement with the reported data [19]. Therefore, it is clear that our

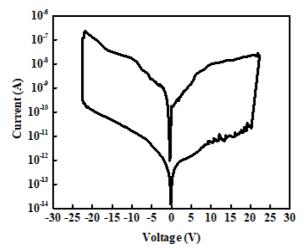


Figure 6. I-V characteristic of fabricated memory device.

CVD grown MoS<sub>2</sub> samples have a single layer with 2H-MoS<sub>2</sub> crystalline structure. We also found that the elemental concentrations of Mo and S are 30.25% and 69.15%, with a ratio of 1.00:2.28. These data further prove the chemical composition of MoS<sub>2</sub>.

Figure 5(a) show the transfer process of the MoS<sub>2</sub> thin films for the fabrication of memory device. PMMA solution was coated by using spin coater. And then, PMMA/MoS<sub>2</sub>/SiO<sub>2</sub>/Si sample was immersed in buffered oxide etch (BOE) solution for etching the SiO<sub>2</sub> layers. After removing the SiO<sub>2</sub> layers, PMMA/MoS<sub>2</sub> structure floating on the BOE solution was deposited on the gold/ SiO<sub>2</sub>/Si substrate. Figure 5(b) displays a schematic diagram of the memory device with Al/PMMA/MoS<sub>2</sub>/Gold structure. In this memory device, gold acts as the electrode and PMMA/MoS<sub>2</sub> structure is used for charges trapping materials.

Figure 6 shows a typical I-V characteristic of the device. The current gradually increased with increasing the applied voltage from 0 to 22 V and then abrupt increase in current from 0.0308 nA to 31.047 nA with a high resistance state/low resistance state ratio up to 10<sup>3</sup>. This memory characteristic of the device is contributed to charge trapping and detrapping between MoS<sub>2</sub> and PMMA. There are highest occupied molecular orbital (HOMO) and lowest unoccupied molecular orbital (LUMO) levels of PMMA with 7.3 eV and 1.8 eV, respectively [20]. MoS<sub>2</sub> has 3.0-4.30 eV electron affinities with a 1.3-1.8 eV band gap. Consequently, PMMA/MoS<sub>2</sub> structures will build a carrier injection barrier as applying voltage. To develop this memory device, a selection of dielectric materials is still required to be optimized which will amplify the I-V characteristic. This PMMA/MoS<sub>2</sub>/gold structure device show efficient memory characteristic. Hence, PMMA/MoS<sub>2</sub>/Metal structure can be used in the next generation non-volatile memory device. However, the further studies on the endurance and retention data should be needed to confirm the usefulness of the 2D material for the resistive switching memory device.

#### **IV. Conclusions**

In summary, we have demonstrated the optimized growing method for MoS<sub>2</sub> single layer. The thin films coverage is highly dependent on the distance between source and substrate. We have successfully grown high surface coverage and large grain size exceeding 80 um by carefully adjusting the distance. We also fabricated the Al/ PMMA/MoS<sub>2</sub>/Metal structure as a resistive sensitive memory device. Typical electrical switching memory effect was observed in this device with a 10<sup>3</sup> current change. The PMMA/MoS<sub>2</sub> structure system can be significant platform for future generation data storage device.

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