# Asymmetric Bond Delta-Polarization at the Interfacial Se—Ru—O Bridge for Efficient pH-Robust Water Electrolysis

Ya Chen, Yaoda Liu, Lei Li, Thangavel Sakthive, Zhixin Guo, and Zhengfei Dai\*

The rationalization of pH-robust catalysis is highly desired but challengeable for overall water electrolysis (WE). It requests a metal active site that can make an efficient adaption with both cathodic hydrogen and anodic oxygen evolution reactions (HER/OER). Herein, a RuO<sub>2-x</sub>/RuSe<sub>2</sub> heterostructure electrocatalyst is profiled with interfacial Se-Ru-O bridge for the pH-robust water splitting studies. An asymmetric bond delta-polarization ( $\Delta p$ ) is found at the interfacial Se–Ru–O bridge, including the  $\Delta p > 0$  at the Ru–O part and  $\Delta p < 0$  at the Ru–Se side by both experiment and calculation results. The enlarged Ru–O bond polarizability ( $\Delta p > 0$ ) can in principle trigger the lattice oxygen mediated (LOM) pathway for OER; meanwhile, the reduced Ru-Se bond polarizability can benefit the HER due to the strengthened d-p band hybridization. Resultantly, the heterostructure can deliver ultralow overpotentials of 25/10 mV for Pt-beyond HER and 210/255 mV for OER at 10 mA cm $^{-2}$  in acidic/alkaline media, respectively. In especial, the acidic overall WE can be stably operated for 200 h with a low cell voltage of 1.478 V at 10 mA cm<sup>-2</sup>. This research clarifies the asymmetric bond polarization as the criterion for the rational design of efficient WE catalysts.

## 1. Introduction

Hydrogen is recognized as a preferential candidate to realize pollution-free and sustainable energy conversion protocols.<sup>[1]</sup> Among various techniques, water electrolysis (WE) promises the ecofriendly manner for hydrogen production, but it should be kinetically accelerated by efficient catalysts.<sup>[2]</sup> For the cathodic hydrogen evolution reaction (HER), Pt-group metals (PGMs) are regarded as the benchmarks due to its high electrocatalytic activity and stability.<sup>[3]</sup> Among the PGMs, Ruthenium with lower price and intense H affinitive capacity (≈65 kcal·mol<sup>-1</sup>) has been deemed as an alternative substitute for Pt benchmark.<sup>[4]</sup> While

Y. Chen, Y. Liu, L. Li, Z. Guo, Z. Dai State Key Laboratory for Mechanical Behavior of Materials Xi'an Jiaotong University Xi'an 710049, P. R. China E-mail: sensdai@mail.xjtu.edu.cn T. Sakthive Department of Chemical Engineering Kumoh National Institute of Technology Gyeongbuk 39177, South Korea

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from Sabatier's principle, it still necessitates the Ru-based catalysts to a moderate H affinity with near-neutral-adsorption dynamics ( $\Delta G_{H^*} \approx 0$ ).<sup>[5]</sup> It is indicated that the introduction of non-metallic elements (NME, Se, S, etc.) can offer an effective measure to balance the hydrogen adsorption/desorption behavior for higher HER activity.<sup>[6]</sup> The bond polarization between them will lighten the strong H-Ru adsorption and induce a Ru/NME dualsite boosted HER kinetics through the d/p band center regulation.<sup>[7]</sup> Although the Ru-compounds currently achieved excellent properties for HER, the extensive WE utilization is significantly impeded by the sluggish anodic multi-step oxygen evolution reaction (OER).<sup>[8]</sup> Besides, the practical WE devices appeals for the electrocatalysts that can tolerate the pH-fluctuation for longterm operation.<sup>[9]</sup> Hence, there is a pressing need for an efficient and pH-robust Rubased catalyst capable of accelerating HER kinetics while also exhibiting OER activity.

For the anode-side OER process, it still remains a bottleneck limitation in acidic condition due to the severe corrosion issue.<sup>[10]</sup> Currently, PGM oxides (e.g., RuO<sub>2</sub>) have gained ever-increasing attention for their corrosion-resistive acidic OER properties.[11] While in the OER activity promotion, it is always confined by the Sabatier's adsorbate evolution mechanism (AEM) with the scaling relationship limitation.<sup>[12]</sup> The community has thus proposed an alternative lattice oxygen mediated (LOM) pathway to overpass this scaling limitation, in which the OER process can be promoted by direct O<sub>Lattice</sub>-\*O coupling.<sup>[13]</sup> But this in turn leads to an excessive oxidation of Ru4+ sites to acidicsoluble  $\operatorname{Ru}^{(4+\delta)+}$  species toward a deteriorated OER durability.<sup>[10a]</sup> The Ru-O bond modulation should be the fundamental in the design of stable and active RuO<sub>2</sub> catalysts.<sup>[14]</sup> To alleviate the  $Ru^{4+} \rightarrow Ru^{(4+\delta)+}$  route, it will be feasible to introduce lowvalence Ru < 4+ sites in  $RuO_2$  to block its direct-oxidation into soluble  $\operatorname{Ru}^{(4+\delta)+}$ . Meanwhile, the presence of low-valence Ru < 4+ sites will also enlarge the Ru–O bond polarizability and weaken the lattice oxygen confinement to benefit the LOM pathway.<sup>[15]</sup> But for the e<sup>-</sup>-accepting HER, a reduced Ru-bond polarizability is demanded to raise the e-cloud overlap between the Ru and non-metal element sites for proton adsorption.<sup>[16]</sup> Hence, asymmetric bond polarization at the Ru sites is expected to actively couple the HER and OER for pH-robust water electrolysis.

> In this study, we have profiled a  $RuO_{2-x}/RuSe_2$  heterostructure as an effective platform for the pH-robust WE performance and catalytic mechanism studies. The electronspun RuO<sub>2x</sub>/RuSe<sub>2</sub> nanofibers is featured with the diameter of ca. 100 nm and viahole porous structure. It is revealed that the RuO<sub>2,v</sub>/RuSe<sub>2</sub> heterostructure undertakes an asymmetric bond delta-polarization  $(\Delta p)$  at the formed interfacial Se–Ru–O bridge. The enlarged Ru–O bond polarizability ( $\Delta p > 0$ ) can in principle activate the lattice oxygen mediated (LOM) pathway for the boosted OER process. Meanwhile, the decreased Ru-Se bond polarizability (Ap < 0) can benefit the HER kinetics due to the enhanced  $d_{Ru}$ -p<sub>Se</sub> band hybridization. As expected, the optimized RuO<sub>2,x</sub>/RuSe<sub>2</sub> heterostructure exhibits the efficient HER/OER catalytic activity and stability in both acidic and alkaline media (25/210 mV in 0.5 м H<sub>2</sub>SO<sub>4</sub>, 10/255 mV in 1 м KOH, at 10 mA cm<sup>-2</sup>). Especially, the overall WE can be stably delivered for 200 h with ultralow operation cell voltages of 1.478 and 1.496 V to reach 10 mA  $\rm cm^{-2}$ in 0.5 M H<sub>2</sub>SO<sub>4</sub> and 1 M KOH, respectively. This work opens up the interesting possibilities for the rational design of efficient bifunctional WE electrocatalysts through bond polarization engineering.

#### 2. Results and Discussion

#### 2.1. Morphology, Phase, and Surface Chemical State Studies

Figure 1a depicts the synthesis process of hollow carbon wrapped RuO<sub>2-x</sub>/RuSe<sub>2</sub> heterostructure (RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF) by a facile electrospinning/calcination process. The detailed synthetic procedures are described in the supplementary information. Figure 1b presents the scanning electron microscope (SEM) image of the as-obtained RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF with an intact fibrous hollow structure. The Brunauer-Emmett-Teller (BET) surface area is determined to be 71.40  $m^2 g^{-1}$ (Figure S1, Supporting Information).<sup>[17]</sup> Such a conductive porous skeleton would merit the active site exposure, electron transport, and bubble release in the electrochemical catalysis.<sup>[18]</sup> The similar morphological features of RuO<sub>2</sub>/RuSe<sub>2</sub>@CNF fabricated with different Se source amount are also depicted in Figure S2 (Supporting Information). The microstructures of the fibrous RuO<sub>2,v</sub>/RuSe<sub>2</sub>@CNF were characterized by transmission electron microscopy (TEM, Figure 1c-e; Figure S3, Supporting Information). The well-defined RuO<sub>2-x</sub>/RuSe<sub>2</sub> nanoparticles (NPs) in carbon matrix can not only prevent their detachment and selfaggregation but also ensure the electronic conductivity.<sup>[19]</sup> The average NP size was determined as ca.  $\approx 25.4$  nm (Figure S4). In the high-resolution TEM (HRTEM, Figure 1d; Figure 55, Supporting Information) images, the *d*-spacing of 0.320 and 0.299 nm can be respectively indexed as the (110) plane of tetragonal RuO<sub>2</sub> and (200) plane of cubic RuSe<sub>2</sub>.<sup>[20]</sup> Corresponding elemental mapping image further confirms the uniform distribution of Ru, Se, C, O, and N elements throughout the fibrous heterostructure (Figure 1e). The Ru content in the  $RuO_{2-x}/RuSe_2@CNF$  catalyst was determined to be ≈8.2 wt% (Table S1, Supporting Information) by inductively coupled plasma-mass spectrometry (ICP-MS) test.

X-ray diffraction (XRD) was applied to examine the phase information of the  $RuO_2/RuSe_2@CNF$  heterostructure and other control samples (Figure 1f; Figure S6, Supporting In-

formation). For RuO2-x/RuSe2@CNF, the typical XRD peaks located at 27.96°/35.08°/40.21°/54.27° correspond to the (110)/(101)/(200)/(211) planes of RuO<sub>2</sub> (JCPDS No. 43–1027);<sup>[21]</sup> and the peaks at 30.06°, 33.68°, 37.10°, 43.05°, 50.95°, 55.74°, 58.08° can be assigned to the (200), (210), (211), (220), (311), (230), and (321) planes of RuSe<sub>2</sub> (JCPDS No. 03-1198).<sup>[6c,22]</sup> In addition, the structures of RuO2, @CNF and RuSe2@CNF were also characterized for reference in Figures S7 and S8 (Supporting Information). All the above results suggest the successful formation of RuO2-x/RuSe2 hetero-interfaces in the porous CNF matrix. The surface states were further checked by Raman, electron paramagnetic resonance (EPR), and Xray photoelectron spectroscopy (XPS). Figure 1g displays the Raman spectra of RuSe2@CNF, RuO2x/RuSe2@CNF, and RuO<sub>2-x</sub>@CNF. For the RuO<sub>2-x</sub>@CNF sample, three distinctive peaks at 515.3, 631.1, and 693.5 cm<sup>-1</sup> can be ascribed to the  $E_{\rm g}$ ,  $A_{1\rm g}$ , and  $B_{2\rm g}$  vibration modes of RuO<sub>2</sub>, respectively.<sup>[20]</sup> Compared to RuO2-x@CNF, the RuO2-B2g peak intensity of RuO<sub>2-v</sub>/RuSe<sub>2</sub>@CNF obviously declined due to the surface selenization. With the oxygen-extracted selenization, more oxygen vacancies can be found in the RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF, as reflected by the slightly increased EPR signal at g = 2.005 (Figure 1h).<sup>[11a]</sup> The Fourier transform infrared spectroscopy (FT-IR) in Figure S9 (Supporting Information) shows that the RuO2-x/RuSe2@CNF possesses a higher O-H vibration intensity than RuO<sub>2,x</sub>@CNF and RuSe2@CNF, indicating better hydrophilicity.<sup>[23]</sup> Besides, the RuO2-x/RuSe2@CNF shows the smallest contract angle (CA, 11.5°) in comparison with RuO2-x@CNF and RuSe<sub>2</sub>@CNF (Figure S10, Supporting Information), implying the higher hydrophilicity and more efficient water adsorption.<sup>[24]</sup>

The chemical states of these samples were further investigated by XPS in Figure 1i-k and Figure S11 (Supporting Information). As depicted in Figure 1i, the high-resolution Ru 3p spectrum of RuO<sub>2-x</sub>@CNF exhibits two typical characteristic peaks centered at 463.36 and 485.60 eV, attributed to the Ru 3p<sub>3/2</sub> and Ru 3p<sub>1/2</sub> of Ru–O bond in RuO<sub>2</sub>, respectively.<sup>[25]</sup> After selenization treatment, two additional deconvoluted peaks at 461.60 and 483.87 eV can be assigned to the Ru  $3p_{3/2}$  and Ru  $3p_{1/2}$  of Ru–Se bond in RuSe<sub>2</sub>, manifesting the formation of the Se-Ru-O link in the heterostructure.<sup>[6c]</sup> Besides, the Se 3d XPS spectrum for RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF (Figure 1j) splits into three characteristic peaks, including Se  $3d_{5/2}$  (55.37 eV), Se  $3d_{3/2}$ (56.21 eV), and SeO<sub>x</sub> (59.01 eV).<sup>[26]</sup> Compared with RuSe<sub>2</sub>@CNF, the Se 3d binding energy of RuO2-x/RuSe2@CNF exhibits an obvious positive-shift about 0.8 eV, indicating the  $RuSe_2 \rightarrow RuO_{2,v}$ electron transfer direction in the heterostructure. This electron transfer manner can be also reflected by the O1s XPS results in Figure 1k, where the RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF peaks are slightly redshifted relative to RuO<sub>2-x</sub>@CNF. Such a selenide-to-oxide interfacial electron transfer behavior can be also understood by their distinctive Se/O electronegativity difference (Figure S12, Supporting Information).<sup>[27]</sup> From the O 1s XPS data, the heterostructure shows the higher contents of oxygen vacancies (530.06 eV) and surface hydroxyl group (531.93 eV),<sup>[28]</sup> in accordance with the EPR and FT-IR/CA results. The strong interfacial interactions will tune the whole electronic structure of the heterostructure, and the enhanced hydrophilicity can benefit the water adsorption.

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**Figure 1.** Synthesis scheme and structural characterizations. a) Schematic diagram for the synthesis of  $RuO_{2\cdot x}/RuSe_2@CNF$ . b) SEM image of  $RuO_{2\cdot x}/RuSe_2@CNF$  (inset: cross-section SEM image). c) TEM image, d) HR-TEM image, and e) elemental mapping of  $RuO_{2\cdot x}/RuSe_2@CNF$ . f) XRD pattern, g) Raman spectra, and h) EPR spectra of  $RuO_{2\cdot x}@CNF$ ,  $RuO_{2\cdot x}/RuSe_2@CNF$ , and  $RuSe_2@CNF$ . i) Ru 3p, j) Se 3d, and k) O 1s XPS spectra of  $RuO_{2\cdot x}@CNF$ ,  $RuO_{2\cdot x}/RuSe_2@CNF$ ,  $RuO_{2\cdot x}/RuSe_2@CNF$ , and  $RuSe_2@CNF$ , and  $RuSe_2@CNF$ , and  $RuSe_2@CNF$ . ii) Ru 3p, j) Se 3d, and k) O 1s XPS spectra of  $RuO_{2\cdot x}@CNF$ ,  $RuO_{2\cdot x}/RuSe_2@CNF$ . ii) Ru 3p, j) Se 3d, and k) O 1s XPS spectra of  $RuO_{2\cdot x}@CNF$ . Functional results are constructed as the second results are

#### 2.2. Coordination Environment and Bond Polarization

X-ray absorption near-edge structure (XANES) and Fouriertransformed EXAFS (FT-EXAFS) were further conducted to investigate the Ru chemical environment of these materials. As presented in **Figure 2**a, the white-line of Ru K-edge XANES for  $RuO_{2x}/RuSe_2@CNF$  is located between Ru foil and  $RuO_2$ . It reveals that the average Ru oxidation state in the heterostructure is between  $Ru^0$  and  $Ru^{4+}$ .<sup>[29]</sup> The fitted Ru valence state in the  $RuO_{2x}/RuSe_2@CNF$  is of ca. +2.28 between the  $RuO_{2x}@CNF$  (+3.32) and  $RuSe_2@CNF$  (+0.85), as shown in Figure 2b. This also suggests the  $RuO_{2x}$  and  $RuSe_2$  behave the electronic

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**Figure 2.** EXAFS spectra and analyses. a) Ru K-edge XANES, b) Average oxidation states of Ru from XANES spectra, c) Ru K-edge FT-EXAFS, d) Ru K-edge EXAFS oscillation function, and e) Wavelet transform for  $k^3$ -weighted EXAFS signal of RuO<sub>2-x</sub>@CNF, RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF, and RuSe<sub>2</sub>@CNF. f) The effects of Ru–O bond length/valence on bond polarizability. g) The delta-bond polarizability ( $\Delta p$ ) at Se–Ru–O interfacial bridge and its effect on OER/HER catalysis.

acceptor and donor in the heterostructure, respectively. Figure 2c presents the corresponding FT-EXAFS spectra of the different samples. For the RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF, two scattering peaks at 1.54 and 2.14 Å are ascribable to the Ru–O and Ru–Se coordination.<sup>[6a,10a]</sup> Relative to the RuO<sub>2-x</sub>@CNF, the Ru–O bond is slightly stretched in RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF heterostructure (1.48  $\rightarrow$  1.54 Å), illustrating a stretched Ru–O bond after RuO<sub>2-x</sub>-RuSe<sub>2</sub> incorporation. The Ru K-edge oscillation curves in Figure 2d also indicate the different local atomic arrangements and structural distortion of the Ru sites in the heterostructure. which is vividly reflected by the wavelet transform EXAFS (WT-EXAFS) analyses in Figure 2e and Figure S13 (Supporting Information). Based on the above XPS and XAFS results, the strong electronic interactions between RuO<sub>2-x</sub> and RuSe<sub>2</sub> is configured via the asymmetric Se–Ru–O interfacial bridge, ac-

celerating the charge transport from Se sites to Ru-bound O (Figure 2f).<sup>[28a]</sup>

Moreover, the bond configuration change will impact the interactions between the catalysts and adsorbates to regulate the electrocatalytic properties. According to Coulomb's law ( $F \propto Qq/r^2$ ), the bond strength (F) is highly related to atomic valence charge (Q, q) and bond length and bond length r (Figure S14, Supporting Information).<sup>[30]</sup> Generally, the bond-length stretch and valence decrease will result in a weaker bond strength, and thus bring an increased bond polarizability ( $\Delta p > 0$ ).<sup>[15]</sup> Here in the Se–Ru–O bridge, the Ru–O half edge undergoes the bond stretching and valence decrease (+3.32  $\rightarrow$  +2.28) relative to the RuO<sub>2-x</sub>, giving rise to an increased bond polarizability ( $\Delta p > 0$ ); while for Ru–Se half edge in the bridge, it concludes a decreased bond polarizability ( $\Delta p < 0$ ) against the RuSe<sub>2</sub>. Therefore, a Janus







**Figure 3.** Electrocatalytic HER performances. a) LSV curves, b) Tafel slopes, and c) histogram of  $\eta_{10}$ -Tafel slope values of different samples in 0.5 M H<sub>2</sub>SO<sub>4</sub>. d) LSV curves, e) Tafel plots, and f) of the HER performance comparison of RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF with reported catalysts in 1 m KOH. g) Mass activities of different catalysts. h) i-t curves of RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF for HER in acidic and alkaline electrolytes. i) HER LSV curves for RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF after 1000th and 3000th cycles in 1 m KOH.

asymmetric bond polarization occurs at the Se–Ru–O bridge in the heterostructure (Figure 2g), which might merit the bifunctional water splitting process as follows: 1) for the Ru–O bond side with  $\Delta p > 0$ , a higher bond polarizability can alleviate the lattice oxygen confinement to trigger the preferable LOM pathway and break the OER scaling-relationship limitation; 2) for the Ru-Se bond side with  $\Delta p < 0$ , the intensified Ru–Se bond strengthens the d-p band hybridization to balance the H<sup>\*</sup> adsorption/desorption on the catalyst surface to benefit the HER kinetics.<sup>[15]</sup>

#### 2.3. Electrocatalytic Properties for Water Splitting

The HER behaviors of different catalysts were studied in both the acidic and alkaline electrolytes by a standard three-electrode system, as shown in **Figure 3**.<sup>[31]</sup> The RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF material is found with superior HER activity and kinetics with a low overpotential ( $\eta_{10}$ ) of 25 mV at 10 mA cm<sup>-2</sup> (Figure 3a) and a

small Tafel slope of 32.1 mV dec<sup>-1</sup> (Figure 3b) in 0.5 M H<sub>2</sub>SO<sub>4</sub>. By contrast (Figure 3c), other control catalysts and even the Pt/C benchmark all show higher  $\eta_{10}$  and hysteretic dynamics, such as RuO<sub>2-x</sub>@CNF (76.2 mV, 63 mV dec<sup>-1</sup>), RuSe<sub>2</sub>@CNF (44.8 mV, 49.4 mV dec<sup>-1</sup>), commercial Ru/C (84.1 mV, 99.8 mV dec<sup>-1</sup>), and Pt/C (33.4 mV, 55.9 mV dec<sup>-1</sup>). As compared with other precious metals catalysts, the RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF material also shows an outperformed acidic HER properties (Table S2, Supporting Information). The electrochemical active surface area (ECSA) of different catalysts are also investigated in Figure S15 (Supporting Information), demonstrating the higher intrinsic HER activity of RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF.<sup>[5]</sup> Besides, electrochemical impedance spectrometry (EIS) measurements indicate the boosted charge transport in the RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF (Figure S16, Supporting Information).<sup>[32]</sup>

Further, the HER activity of each catalyst was also evaluated in 1 mm KOH electrolyte. It is found that the RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF catalyst delivers a Pt-beyond  $\eta_{10}$  overpotential of 10 mV and low Tafel slope of 37.6 mV dec<sup>-1</sup> in 1 mmm KOH (Figure 3d,e). Such an alkaline HER performance is far superior to Ru/C, Pt/C (19.1 mV, 53.7 mV dec<sup>-1</sup>), and other controls (Figure S17, Supporting Information) as well as the state-of-the-art Ru-based HER catalysts (Figure 3f; Table S3, Supporting Information). The ECSA and EIS results further confirm that the RuO2x/RuSe2@CNF processes the optimized intrinsic catalytic activity and electrontransfer characteristics among all the catalysts in alkaline HER (Figures \$18 and \$19, Supporting Information). In addition, the mass activities of RuO2-x/RuSe2@CNF in alkaline (Figure 3g) and acidic (Figure S20, Supporting Information) media were also calculated. The RuO2-x/RuSe2@CNF catalyst presents Ptbeyond HER mass activities in both acidic and alkaline media. The HER stability of the RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF were further explored through i-t tests, as shown in Figure 3h. It clearly displays that the HER rate activity (20 mA cm<sup>-2</sup>) can be maintained well without significant attenuation for 140 h in both 0.5 M H<sub>2</sub>SO<sub>4</sub> and 1 м KOH. Figure S21 (Supporting Information) further shows the stable HER catalytic performance of  $RuO_{2-x}/RuSe_2@CNF$  under a higher current density of 100 mA cm<sup>-2</sup>. The post-mortem SEM, TEM, XRD, and XPS examinations indicate the retention of RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF structure under acidic/alkaline HER conditions (Figures S22-S24, Supporting Information). The pHrobust HER stability of RuO2.x/RuSe2@CNF can be also reflected by the cycling tests in 1 M KOH (Figure 3i) and 0.5 M H<sub>2</sub>SO<sub>4</sub> (Figure S25, Supporting Information), where the currents after 3000 cycles show negligible decays. The promoted HER property can be understood by the decreased Ru-Se bond polarizability at the Se-Ru-O bridge with more balanced the H\* adsorption/desorption.

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The OER performance of each catalysts was also evaluated in both acidic and alkaline media. As presented in **Figure 4**a,b, the  $RuO_{2,x}/RuSe_2@CNF$  catalyst delivers an ultralow  $\eta_{10}$  overpotential of 210 mV with fast kinetics (43.3 mV dec<sup>-1</sup>) in 0.5 M H<sub>2</sub>SO<sub>4</sub>. As compared in Figure S26a (Supporting Information), the acidic OER performance of RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF is much superior to RuO<sub>2-x</sub>@CNF (245 mV, 65.6 mV dec<sup>-1</sup>), RuSe<sub>2</sub>@CNF (400 mV, 510.8 mV dec<sup>-1</sup>), and commercial IrO<sub>2</sub> (270 mV, 83.8 mV dec<sup>-1</sup>). Importantly, the RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF exceeds most reported Rubased acidic OER electrocatalysts in terms of  $\eta_{10}$  and Tafel slopes (Figure 4c; Table S4, Supporting Information). Figures S26b and S27 (Supporting Information) present the alkaline OER properties of different catalysts, where the heterostructure catalyst still displays the better alkaline OER performances (255 mV, 52.5 mV dec<sup>-1</sup>) than other control catalysts and even the commercial benchmarks. The ECSA results in Figure S28 (Supporting Information) suggest that the heterostructure catalyst is endowed with a higher intrinsic alkaline OER activity. Overall, the RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF catalyst has presented the optimal OER activities in both acidic and alkaline conditions (Figure S29, Supporting Information).

As for the OER durability, the  $\text{RuO}_{2\cdot x}/\text{RuSe}_2$ @CNF can stably maintain the current density for 140 h and 3000 times cycling without obvious decays in both acidic (Figure 4d,e) and alkaline (Figure S30, Supporting Information) electrolytes. On the contrary, the  $\text{RuO}_{2\cdot x}$ @CNF and commercial  $\text{RuO}_2$  catalysts show the sharply dropped current behaviors in acidic OER processes (Figure 4d). The enhanced acidic OER stability was further illustrated by inductively coupled plasma optical emission spectrometer (ICP-OES, Figure 4f).<sup>[10a,33]</sup> The dissolved Ru amount

in RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF is only 1/10 of the RuO<sub>2-x</sub>@CNF during the acidic OER, indicating the hindered corrosion of RuO<sub>2,v</sub> after RuSe<sub>2</sub> confinement. Moreover, the post-OER TEM characterizations have demonstrated the well-persisted morphology and microstructure of the RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF (Figure 4e inset; Figures S31 and S32, Supporting Information). XRD and XPS were employed to check the RuO<sub>2,x</sub>/RuSe<sub>2</sub>@CNF after OER test. The attenuated Ru-Se content during the OER may be attributed to the anodic oxidation and the dissolution of Se element (Figures S33 and S34, Supporting Information). This also manifests the catalytic role of Ru-O content and the protective effect of Ru-Se content for OER in the heterostructure system. In addition, we have also checked the effect of selenization degree (RuO<sub>2</sub>/Se source ratio) on the HER/OER performances in both acidic and alkaline media (Figure \$35, Supporting Information). It is suggested that a moderate selenization can make a fit between the protective and catalytic effects for the OER promotion.

Encouraged by the pH-robust HER/OER properties, the RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF catalyst was further examined in symmetric overall water splitting (OWS) under acidic and alkaline media (Figure 4g-i). The RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF||RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF couple shows the much superior OWS performances to those commercial Pt/C||RuO<sub>2</sub> and Pt/C||IrO<sub>2</sub> pairs. It just requires ultralow cell voltages of 1.496 V (Figure 4g) and 1.478 V (Figure 4h) at 10 mA cm<sup>-2</sup> to deliver the OWS in 1 M KOH and 0.5 M H<sub>2</sub>SO<sub>4</sub>, respectively. A solar-powered (1.48 V) OWS device was also assembled to demonstrate the effectiveness in the sustainable H<sub>2</sub> production (Figure 4h inset). In both alkaline and acidic electrolytes, the RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF electrode can deliver a high OWS activity with a negligible attenuation rate for 200 h at 20 mA  $cm^{-2}$  (Figure 4i). It is worth noting that the RuO<sub>2,v</sub>/RuSe<sub>2</sub>@CNF material is among the best list of noble-metal-based OWS electrocatalysts (Figure 4j; Table S5, Supporting Information). Figure S36 (Supporting Information) presents the Faradaic efficiency (FE) of over 98.8% for the RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF electrodes toward highly efficient OWS.<sup>[34]</sup>

#### 2.4. Enhanced Mechanism Analyses

Density functional theory (DFT) calculations were employed to gain an insight into the promoted catalytic properties of the RuO<sub>2-x</sub>/RuSe<sub>2</sub> heterostructure. Figure 5a shows the heterostructure model, and the counterpart models of RuO2-x and RuSe2 are presented in Figures S37 and S38 (Supporting Information). At the  $RuO_{2,x}/RuSe_2$  interface, it is suggested that the electron transfers from  $RuSe_2$  to  $RuO_{2-x}$  by the difference charge density simulations (Figure S39, Supporting Information). The electron localization function (ELF) map in Figure 5b further confirms the presence of strong electron coupling at the interface through the asymmetric Se-Ru-O configuration. Such a charge transport pathway was also validated by the result of their work function difference (Figure 5c). Given that water adsorption is a prerequisite step for water splitting, we further calculated the H<sub>2</sub>O adsorption energy ( $\Delta E_{H2O}$ ) on RuO<sub>2-x</sub>, RuO<sub>2-x</sub>/RuSe<sub>2</sub>, and RuSe<sub>2</sub> systems. The RuO<sub>2-x</sub>/RuSe<sub>2</sub> exhibits higher  $\Delta E_{H2O}$  (-1.149 eV) than the  $\text{RuO}_{2\text{-x}}$  (–0.564 eV) and  $\text{RuSe}_2$  (–0.729 eV), presenting a more hydrophilic surface to benefit subsequent HER/OER catalytic steps (Figure 5d; Table S6, Supporting Information). For

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**Figure 4.** OER and OWS performances. a) OER polarization curves, b) Tafel slopes, and c) OER performance comparison of  $RuO_{2,x}/RuSe_2@CNF$  with other catalysts in 0.5 M  $H_2SO_4$ . d) I-t curves of  $RuO_{2,x}/RuSe_2@CNF$ ,  $RuO_{2,x}@CNF$  and commercial  $RuO_2$  at 20 mA cm<sup>-2</sup> in 0.5 M  $H_2SO_4$ . e) OER LSV curves for  $RuO_{2,x}/RuSe_2@CNF$  after 1000th and 3000th cycles in 0.5 M  $H_2SO_4$ , inset is the post-mortem HRTEM image. f) Dissolved Ru amount during OER in 0.5 M  $H_2SO_4$  by ICP-OES for the  $RuO_{2,x}/RuSe_2@CNF$  and  $RuO_{2,x}@CNF$ . OWS polarization curves of  $RuO_{2,x}/RuSe_2@CNF$  and controls in g) 0.5 M  $H_2SO_4$  and h) 1 M KOH. i) I-t tests of  $RuO_{2,x}/RuSe_2@CNF ||RuO_{2,x}/RuSe_2@CNF$  at 10 mA cm<sup>-2</sup> in acidic and alkaline electrolytes for 200 h. j) Performance comparison of  $RuO_{2,x}/RuSe_2@CNF$  with recently excellent catalysts for OWS in acidic and alkaline electrolytes, respectively.

the subsequent H<sub>2</sub>O dissociation process, the RuO<sub>2.x</sub>/RuSe<sub>2</sub> displayed the lowest H<sub>2</sub>O dissociation energy (0.36 eV), suggesting the facilitated H<sub>2</sub>O dissociation and H<sup>\*</sup> intermediates formation behavior (Figure S40, Supporting Information). As for HER, the hydrogen adsorption free energy ( $\Delta G_{H^*}$ ) was investigated for

RuO<sub>2-x</sub>, RuO<sub>2-x</sub>/RuSe<sub>2</sub>, and RuSe<sub>2</sub> catalysts.<sup>[35]</sup> From the step energy diagrams in Figure 5e, the Ru<sub>@Ru-Se</sub> site in the heterostructure, as the HER active center, displays the more neutral  $\Delta G_{H^*}$  value (-0.053 eV) than those of Ru site in RuO<sub>2-x</sub> and the Ru/Se sites in RuSe<sub>2</sub>. Corresponding atomic adsorption states and

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**Figure 5.** Catalytic enhancement mechanism analyses. a) Structure model and b) electron localization function of  $RuO_{2\cdot x}/RuSe_2$  heterostructure. c) Work function diagram between  $RuO_{2\cdot x}$  and  $RuSe_2$ . d)  $\Delta E_{H2O}$  on the  $RuO_{2\cdot x}$ ,  $RuO_{2\cdot x}/RuSe_2$ , and  $RuSe_2$  surface. e) Energy barriers for HER at different sites. f) OER free energy diagrams for  $RuO_{2\cdot x}/RuSe_2$ , and  $RuO_{2\cdot x}$ . g) OER LSV curves of  $RuO_{2\cdot x}/RuSe_2$ @CNF in electrolytes with different pH values. h) Current densities of  $RuO_{2\cdot x}/RuSe_2$ @CNF and  $RuO_{2\cdot x}/RuSe_2$ @CNF at 1.60 V versus RHE against the pH values. i) LSV curves of  $RuO_{2\cdot x}/RuSe_2$ @CNF and  $RuO_{2\cdot x}/RuSe_2$ @CNF and  $RuO_{2\cdot x}/RuSe_2$ @CNF in 1 M KOH and 1 M TMAOH. j) OER energy barriers via AEM and LOM pathways on  $RuO_{2\cdot x}/RuSe_2$ . k) The PDOSs projected on the O-p and Ru-d orbits for  $RuO_{2\cdot x}/RuSe_2$  and  $RuO_{2\cdot x}$ .

calculated energy value data are presented in Figures S41 and S42 and Table S7 (Supporting Information). Such a neutral  $\Delta G_{H^*}$  in the heterostructure is responsive for balancing the hydrogen ads-/de-sorption behaviors toward boosted HER performance.

Moreover, the OER pathway was investigated to decipher the OER catalytic enhancement mechanism. First of all, the conventional adsorbate evolution mechanism (AEM) was applied to discern the catalytic activity difference of  $RuO_{2-x}/RuSe_2$  and  $RuO_{2-x}$  (Figure 5f; Figure S43a, Supporting Information). The



adsorption sites of the intermediates (\*OH, \*O, and \*OOH) on each system are shown in Figures S44–S47 (Supporting Information), and Table S8 (Supporting Information) listed the corresponding calculated theoretical overpotential ( $\eta$ ) values. It is revealed that both the rate-determining steps (RDS) of RuO<sub>2-x</sub> and RuO<sub>2-x</sub>/RuSe<sub>2</sub> are the evolution from \*O to \*OOH (\*O  $\rightarrow$  \*OOH step) intermediates (Figure 5f). The RuO<sub>2-x</sub>/RuSe<sub>2</sub> undergoes more balanced RDS energy barriers (1.67 eV@0 V and 0.45 eV@1.23 V) than that of RuO<sub>2-x</sub> (2.14 eV@0 V and 0.91 eV@1.23 V), thereby strengthening the OER activity. Moreover, the RuSe<sub>2</sub> side of RuO<sub>2-x</sub>/RuSe<sub>2</sub> heterostructure was also investigated, and the OER theoretical activities of the Ru/Se sites (RDS: 2.30/2.56 eV@0 V) are both inferior to the Ru site at RuO<sub>2</sub> side (Figures S48 and S49, Supporting Information).

In regard of the high Ru–O bond polarizability ( $\Delta p > 0$ ), the LOM pathway would be the underlying OER manner for the RuO<sub>2-v</sub>/RuSe<sub>2</sub> material.<sup>[24]</sup> Since the LOM pathway features the strong nonconcerted proton-electron coupling,<sup>[13c,36]</sup> the pH-dependent OER activities were investigated for the RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF and RuO<sub>2-x</sub>@CNF in Figure 5g and Figure S50 (Supporting Information). With the pH value increase from 12.5 to 14, the RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF exhibits a higher *j*-pH sensitivity (Figure 5h) than RuO<sub>2-x</sub>@CNF, indicating a boosted OER deprotonation process toward the LOM pathway.<sup>[18,37]</sup> To verify the proposed LOM route, the tetramethylammonium cation (TMA<sup>+</sup>)-poisoning experiments were conducted during the OER (Figure 5i), because the TMA<sup>+</sup> can bind strongly with the  $O_2^{2-}/O^{2-}$  species and hamper the OER-LOM pathway.<sup>[13a,38]</sup> It is observed that the OER activity of RuO2-x/RuSe2@CNF declined significantly in 1 м ТМАОН than that in 1 м КОН. This confirms that the LOM mode is an energetically favorable pathway for the bond-polarized RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF. The free energy diagram of the multistep OER process via AEM and LOM are studied and presented in Figure 5j. The theoretical overpotential  $(\eta)$ of the LOM route (0.38 V) is lower than that of the AEM route (0.45 V) on the RuO<sub>2,x</sub>/RuSe<sub>2</sub> heterostructure. The PDOSs of  $Ru_{4d}$  and  $O_{2p}$  orbits were plotted to further rationalize the promoted OER catalytic activity (Figure 5k). The RuO<sub>2-x</sub>/RuSe<sub>2</sub> heterostructure shows the narrower d-p band center distance (4.23-1.94 = 2.29 eV) than that of RuO<sub>2-x</sub> (4.35–1.73 = 2.62 eV). Such an enhanced d-p band proximity is conductive to the more balanced intermediates (\*OH, \*O, etc.) adsorptions and charge transport for the boosted OER process.<sup>[26]</sup>

## 3. Conclusion

In summary, we have profiled and fabricated a RuO<sub>2-x</sub>/RuSe<sub>2</sub>@CNF heterostructure by a facile electrospinning/selenization process for the efficient and pH-robust WE catalysis. The electronspun RuO<sub>2-x</sub>/RuSe<sub>2</sub> nanofibers is featured with the via-hole porous structure to benefit the electrolyte accessibility. As for HER, the optimized RuO<sub>2-x</sub>/RuSe<sub>2</sub> heterostructure exhibits the Pt-beyond properties with ultralow  $\eta_{10}$  of 10 mV in 1 M KOH and 25 mV in 0.5 M H<sub>2</sub>SO<sub>4</sub>. With respect to OER, the heterostructure can present the competitive performances with low  $\eta_{10}$  of 210 and 255 mV in acidic and alkaline electrolytes, respectively. Furthermore, the overall WE device can be stably delivered for 200 h with ultralow  $\eta_{10}$  cell voltages of 1.478 (acidic) and 1.496 V (alkaline), respectively. It is revealed that the  $RuO_{2\cdot x}/RuSe_2$  heterostructure features an asymmetric bond delta-polarization at the formed interfacial Se–Ru–O bridge: i) the Ru–O bond of  $\Delta p > 0$  triggers the LOM pathway for OER and ii) Ru–Se bond of  $\Delta p < 0$  benefits the HER kinetics by the enhanced  $d_{Ru}$ - $p_{Se}$  band hybridization. This study carries the valuable fundamental to rationalize the pH-robust overall water splitting electrocatalysts through the interface bond modulations.

## **Supporting Information**

Supporting Information is available from the Wiley Online Library or from the author.

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## **Conflict of Interest**

The authors declare no conflict of interest

## **Data Availability Statements**

The data that support the findings of this study are available from the corresponding author upon reasonable request.

## **Keywords**

asymmetric bond-delta polarization, interfacial Se-Ru-O bridge, overall water splitting, pH-robust electrocatalysis, ruthenium oxide

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- a) B. Zhang, J. Wang, G. Liu, C. M. Weiss, D. Liu, Y. Chen, L. Xia, P. Zhou, M. Gao, Y. Liu, W. Sun, *Nat. Catal.* **2024**, *7*, 441; b) R. Detchon, R. Van Leeuwen, *Nature* **2014**, *508*, 309; c) Y. Chen, Y. Liu, W. Zhai, H. Liu, T. Sakthivel, S. Guo, Z. Dai, *Adv. Energy Mater.* **2024**, *14*, 2400059; d) X. He, P. Du, G. Yu, R. Wang, Y. Long, B. Deng, C. Yang, W. Zhao, Z. Zhang, K. Huang, *Small Methods* **2023**, *7*, 2300544.
- [2] a) S. Zhou, H. Jang, Q. Qin, L. Hou, M. G. Kim, S. Liu, X. Liu, J. Cho, *Angew. Chem., Int. Ed.* 2022, *61*, 202212196; b) J. Wang, H. Yang, F. Li, L. Li, J. Wu, S. Liu, T. Cheng, Y. Xu, Q. Shao, X. Huang, *Sci. Adv.* 2022, *8*, eabl9271; c) L. Yang, H. Ren, Q. Liang, K. N. Dinh, R. Dangol, Q. Yan, *Small* 2020, *16*, 1906766.
- [3] a) K. Jiang, B. Liu, M. Luo, S. Ning, M. Peng, Y. Zhao, Y. R. Lu, T. S. Chan, F. M. de Groot, Y. Tan, *Nat. Commun.* **2019**, *10*, 1743; b) H. Zhang, T. Wei, Y. Qiu, S. Zhang, Q. Liu, G. Hu, J. Luo, X. Liu, *Small* **2023**, *19*, 2207249; c) D. Wu, K. Kusada, Y. Nanba, M. Koyama, T. Yamamoto, T. Toriyama, S. Matsumura, O. Seo, I. Gueye, J. Kim, *J. Am. Chem. Soc.* **2022**, *144*, 3365.

#### ADVANCED SCIENCE NEWS

www.advancedsciencenews.com

- [4] a) J. Yu, Q. He, G. Yang, W. Zhou, Z. Shao, M. Ni, ACS Catal. 2019, 9, 9973; b) H. Chen, X. Ai, W. Liu, Z. Xie, W. Feng, W. Chen, X. Zou, Angew. Chem., Int. Ed. 2019, 131, 11531.
- [5] T. Chao, W. Xie, Y. Hu, G. Yu, T. Zhao, C. Chen, Z. Zhang, X. Hong, H. Jin, D. Wang, *Energy Environ. Sci.* 2024, 17, 1397.
- [6] a) K. Wang, J. Zhou, M. Sun, F. Lin, B. Huang, F. Lv, L. Zeng, Q. Zhang, L. Gu, M. Luo, Adv. Mater. 2023, 35, 2300980; b) Z. Pu, I. S. Amiinu, Z. Kou, W. Li, S. Mu, Angew. Chem., Int. Ed. 2017, 56, 11559; c) D. Chen, R. Lu, Y. Yao, D. Wu, H. Zhao, R. Yu, Z. Pu, P. Wang, J. Zhu, J. Yu, J. Mater. Chem. A 2022, 10, 7637; d) K. Huang, C. Lin, G. Yu, P. Du, X. Xie, X. He, Z. Zheng, N. Sun, H. Tang, X. Li, Adv. Funct. Mater. 2023, 33, 2211102; e) P. Du, C. Lin, X. He, Z. Zheng, X. Xie, K. Huang, M. Lei, H. Tang, Adv. Compos. Hybrid Mater. 2023, 6, 40.
- [7] S. Chen, S. Zhang, L. Guo, L. Pan, C. Shi, X. Zhang, Z. F. Huang, G. Yang, J. J. Zou, Nat. Commun. 2023, 14, 4127.
- [8] a) G. Zhang, G. Wang, Y. Liu, H. Liu, J. Qu, J. Li, *J. Am. Chem. Soc.* **2016**, *138*, 14686; b) Y. Liu, X. Li, S. Zhang, Z. Wang, Q. Wang, Y. He,
  W. H. Huang, Q. Sun, X. Zhong, J. Hu, *Adv. Mater.* **2023**, *35*, 2300945;
  c) B. Zhang, Y. Chen, J. Wang, H. Pan, W. Sun, *Adv. Funct. Mater.* **2022**, *32*, 2202227.
- [9] a) C. X. Zhao, J. N. Liu, J. Wang, D. Ren, B. Q. Li, Q. Zhang, *Chem. Soc. Rev.* 2021, *50*, 7745; b) X. Xie, L. Du, L. Yan, S. Park, Y. Qiu, J. Sokolowski, W. Wang, Y. Shao, *Adv. Funct. Mater.* 2022, *32*, 2110036; c) J. Shan, T. Ling, K. Davey, Y. Zheng, S. Z. Qiao, *Adv. Mater.* 2019, *31*, 1900510.
- [10] a) L. Deng, S. F. Hung, Z. Y. Lin, Y. Zhang, C. Zhang, Y. Hao, S. Liu, C. H. Kuo, H. Y. Chen, J. Peng, *Adv. Mater.* **2023**, *35*, 2305939; b) L. An, C. Wei, M. Lu, H. Liu, Y. Chen, G. G. Scherer, A. C. Fisher, P. Xi, Z. J. Xu, C. H. Yan, *Adv. Mater.* **2021**, *33*, 2006328.
- [11] a) H. Yan, Z. Jiang, B. Deng, Y. Wang, Z. J. Jiang, *Adv. Energy Mater.* **2023**, *13*, 2300152; b) Y. Li, W. Wang, M. Cheng, Y. Feng, X. Han, Q. Qian, Y. Zhu, G. Zhang, *Adv. Mater.* **2023**, *35*, 2206351.
- [12] a) Z. Chen, L. Guo, L. Pan, T. Yan, Z. He, Y. Li, C. Shi, Z. F. Huang, X. Zhang, J. J. Zou, *Adv. Energy Mater.* **2022**, *12*, 2103670; b) C. Lin, J. L. Li, X. Li, S. Yang, W. Luo, Y. Zhang, S. H. Kim, D. H. Kim, S. S. Shinde, Y. F. Li, *Nat. Catal.* **2021**, *4*, 1012.
- [13] a) K. Xiao, Y. Wang, P. Wu, L. Hou, Z. Q. Liu, Angew. Chem., Int. Ed.
   2023, 135, 202301408; b) Z. Shi, Y. Wang, J. Li, X. Wang, Y. Wang, Y. Li, W. Xu, Z. Jiang, C. Liu, W. Xing, Joule 2021, 5, 2164; c) A. Grimaud,
   O. Diaz-Morales, B. Han, W. T. Hong, Y. L. Lee, L. Giordano, K. A. Stoerzinger, M. T. Koper, Y. Shao-Horn, Nat. Chem. 2017, 9, 457.
- [14] a) Y. Wang, X. Lei, B. Zhang, B. Bai, P. Das, T. Azam, J. Xiao, Z. S. Wu, Angew. Chem., Int. Ed. 2024, 63, 202316903; b) J. Wang, C. Cheng, Q. Yuan, H. Yang, F. Meng, Q. Zhang, L. Gu, J. Cao, L. Li, S. C. Haw, Chem 2022, 8, 1673.
- [15] W. Zhai, Y. Chen, Y. Liu, T. Sakthivel, Y. Ma, Y. Qin, Y. Qu, Z. Dai, ACS Nano 2023, 17, 17254.
- [16] a) S. Zhao, Y. Liu, Y. Chen, L. Li, W. Zhai, Z. Guo, Z. Dai, J. Mater. Chem. A 2024, 12, 10704 b) L. Deng, F. Hu, M. Ma, S. C. Huang, Y. Xiong, H. Y. Chen, L. Li, S. Peng, Angew. Chem., Int. Ed. 2021, 60, 22276.
- [17] C. Xia, L. Huang, D. Yan, A. I. Douka, W. Guo, K. Qi, B. Y. Xia, Adv. Funct. Mater. 2021, 31, 2105021.
- [18] S. Xin, Y. Tang, B. Jia, Z. Zhang, C. Li, R. Bao, C. Li, J. Yi, J. Wang, T. Ma, Adv. Funct. Mater. 2023, 33, 2305243.

- [19] T. Li, S. Li, Q. Liu, J. Yin, D. Sun, M. Zhang, L. Xu, Y. Tang, Y. Zhang, Adv. Sci. 2020, 7, 1902371.
- [20] Y. Wang, R. Yang, Y. Ding, B. Zhang, H. Li, B. Bai, M. Li, Y. Cui, J. Xiao, Z. S. Wu, Nat. Commun. 2023, 14, 1412;
- [21] C. Hu, K. Yue, J. Han, X. Liu, L. Liu, Q. Liu, Q. Kong, C. W. Pao, Z. Hu, K. Suenaga, *Sci. Adv.* **2023**, *9*, 9144.
- [22] Y. Zhao, H. Cong, P. Li, D. Wu, S. Chen, W. Luo, Angew. Chem., Int. Ed. 2021, 133, 7089.
- [23] Y. Liu, L. Li, L. Wang, N. Li, X. Zhao, Y. Chen, T. Sakthivel, Z. Dai, Nat. Commun. 2024, 15, 2851.
- [24] Q. M. Luo, Y. W. Zhao, L. Sun, C. Wang, H. Q. Xin, J. X. Song, D. Y. Li, F. Ma, Chem. Eng. J. 2022, 437, 135376.
- [25] S. C. Sun, H. Jiang, Z. Y. Chen, Q. Chen, M. Y. Ma, L. Zhen, B. Song, C. Y. Xu, Angew. Chem., Int. Ed. 2022, 61, 202202519.
- [26] a) W. Zhai, Y. Chen, Y. Liu, Y. Ma, P. Vijayakumar, Y. Qin, Y. Qu, Z. Dai, *Nano-Micro Lett.* **2024**, *16*, 115; b) Y. Luo, P. Wang, G. Zhang, S. Wu, Z. Chen, H. Ranganathan, S. Sun, Z. Shi, *Chem. Eng. J.* **2023**, *454*, 140061.
- [27] W. Yan, H. Ma, X. Zhao, Y. Zhang, P. Vishniakov, X. Wang, X. Zhong,
   Z. Hong, M. Y. Maximov, L. Song, *Small* **2023**, *19*, 2208270.
- [28] a) H. Yu, M. Hu, C. Chen, C. Hu, Q. Li, F. Hu, S. Peng, J. Ma, Angew. Chem., Int. Ed. 2023, 62, 202314569; b) Y. Chen, Y. Liu, L. Li, T. Sakthivel, Z. Guo, Z. Dai, Adv. Funct. Mater. 2024, 34, 2401452; c) P. Chen, X. Su, C. Wang, G. Zhang, T. Zhang, G. Xu, L. Chen, Angew. Chem., Int. Ed. 2023, 135, 202306224.
- [29] a) C. Rong, X. Shen, Y. Wang, L. Thomsen, T. Zhao, Y. Li, X. Lu, R. Amal, C. Zhao, *Adv. Mater.* **2022**, *34*, 2110103. b) J. Yang, Y. Shen, Y. Sun, J. Xian, Y. Long, G. Li, *Angew. Chem., Int. Ed.* **2023**, *135*, 202302220.
- [30] L. Zhang, J. Zhang, H. Yu, J. Yu, Adv. Mater. 2022, 34, 2107668.
- [31] a) D. Chen, R. Lu, R. Yu, Y. Dai, H. Zhao, D. Wu, P. Wang, J. Zhu, Z. Pu, L. Chen, Angew. Chem., Int. Ed. 2022, 61, 202208642; b) L. Hou, Z. Li, H. Jang, Y. Wang, X. Cui, X. Gu, M. G. Kim, L. Feng, S. Liu, X. Liu, Adv. Energy Mater. 2023, 13, 2300177.
- [32] H. Xin, L. Sun, Y. Zhao, Q. Luo, S. Guo, D. Li, Y. Chen, N. Ogiwara, H. Kitagawa, *Appl Catal B* **2023**, *330*, 122645.
- [33] a) Y. Xu, Z. Mao, J. Zhang, J. Ji, Y. Zou, M. Dong, B. Fu, M. Hu, K. Zhang, Z. Chen, S. Chen, H. Yin, P. Liu, H. Zhao, *Angew. Chem., Int. Ed.* **2024**, *136*, 202316029; b) Y. Wen, P. Chen, L. Wang, S. Li, Z. Wang, J. Abed, X. Mao, Y. Min, C. T. Dinh, P. D. Luna, R. Huang, L. Zhang, L. Wang, L. Wang, R. J. Nielsen, H. Li, T. Zhuang, C. Ke, O. Voznyy, Y. Hu, Y. Li, W. A. Goddard Iii, B. Zhang, H. Peng, E. H. Sargent, *J. Am. Chem. Soc.* **2021**, *143*, 6482.
- [34] a) R. Liu, M. Sun, X. Liu, Z. Lv, X. Yu, J. Wang, Y. Liu, L. Li, X. Feng,
   W. Yang, Angew. Chem., Int. Ed. 2023, 135, 202312644; b) D. Feng, P.
   Wang, R. Qin, W. Shi, L. Gong, J. Zhu, Q. Ma, L. Chen, J. Yu, S. Liu,
   Adv. Sci. 2023, 10, 2300342.
- [35] Z. W. Seh, J. Kibsgaard, C. F. Dickens, I. Chorkendorff, J. K. Nørskov, T. F. Jaramillo, *Science* 2017, 355, 4998.
- [36] Y. Pan, X. Xu, Y. Zhong, L. Ge, Y. Chen, J.-P. M. Veder, D. Guan, R. O'Hayre, M. Li, G. Wang, *Nat. Commun.* **2020**, *11*, 2002,.
- [37] R. Chen, Z. Wang, S. Chen, W. Wu, Y. Zhu, J. Zhong, N. Cheng, ACS Energy Lett. 2023, 8, 3504.
- [38] X. Chen, Q. Wang, Y. Cheng, H. Xing, J. Li, X. Zhu, L. Ma, Y. Li, D. Liu, Adv. Funct. Mater. 2022, 32, 2112674.

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